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Determination of trace amounts of mercury using hierarchically nanostructured europium oxide

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ABSTRACT

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Keywords: Mercury Europium oxide Fluorescence spectrophotometry Hierarchical nanostructure This work reports a highly sensitive procedure for the determination of trace amounts of mercury, based on fluorescence quenching of thenoyltrifluoroacetone (TTA) capped hierarchically nanostructured europium oxide (cHN-Eu₂O₃). The HN-Eu₂O₃ consisted of nanometer-thick Eu₂O₃ sheets self-organized into nano- and micro-sized monoliths with a hierarchical architecture while retaining its desirable fluorescence properties. The fluorescence intensity of the cHN-Eu₂O₃ was 1000 times higher than that of commercial Eu₂O₃ nanoparticles (equivalent weight) when it was capped with TTA, suggesting that a synergetic effect, confining the longtime Eu³⁺ excitation within the nanostructure and light-harvesting effect of the capping agent, is responsible for this fluorescence enhancement. Excellent interaction between the cHN-Eu₂O₃ and solution species is expected owing to its large surface area, high surface-to-bulk ratio, and ultrahigh fluorescence intensity. As an example, aqueous suspensions of the cHN-Eu₂O₃ were used as sensing agent for the determination of trace amounts of mercury. A linear relationship between the concentration of mercury and fluorescence quenching was observed from 10 ppb to 10 ppm with a correlation coefficient of 0.997 and a detection limit of 5.0 ppb. Mercury in various samples was analyzed using the cHN-Eu₂O₃ suspension.

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1. Introduction

Nanomaterials have attracted tremendous interest as these zero-dimensional nanoparticles [1] and quantum dots [2], onedimensional nanotubes [3], nanowires [4], and nanoribbons [5] exhibit remarkable electronic, magnetic, optical, and catalytic properties that are distinctly different from their bulk counterparts [6,7]. In recent years, there has been a paradigm shift in the field of nanomaterials research, from synthesizing these nanoscale building blocks of arbitrary dimensions and morphologies to constructing three-dimensional (3D) hierarchical nanostructures with controllable architecture and functionality [6–8]. The formation of hierarchical nanostructures is generally considered to be a selfassembly process, in which building blocks self-assemble into regular higher level structures.

Europium oxide (Eu_2O_3) , the stable sesquioxide of Eu^{3+} ion, has been employed extensively as an essential component phosphor in cathode-ray tube displays and as a fluorescent agent in light-emitting compounds [9]. Moreover, Eu_2O_3 is an attractive material for use in a variety of device size regimes, from nanoscale to macroscale due to its attractive fluorescence properties such as long emission lifetime, large Stokes shift and ultranarrow emission peaks [10-13]. Recent studies have sought to take advantage of those optical properties of Eu₂O₃ for implementation in nanoscale optical devices, such as nanocrystalline light-emitting diodes and biosensing agents [14,15]. We report herein, for the first time, the preparation of hierarchical nanostructures of Eu₂O₃ (HN-Eu₂O₃). These nanostructures consisted of nanometer-thick Eu₂O₃ sheets self-organized into nano- to micro-sized monoliths with a hierarchical architecture. The product obtained retained its desirable fluorescence properties that are typical of europium-based materials [9,16]. Unlike other fluorescent materials, for example quantum dots [17], the fluorescence properties of the HN-Eu₂O₃ originate from Eu³⁺ ion absorption are relatively unaffected by the size or the surface morphology of the material. Capping the HN-Eu₂O₃ with thenoyltrifluoroacetone (TTA) turned it into an ultrabright fluorescence sphere, noticeably visible under a fluorescence microscope. The cHN-Eu₂O₃ is of particular interest for applications in catalysis, energy conversion and storage, environmental abatement, and sensors. Attempts were made in applying the cHN-Eu₂O₃ in the determination of traces of mercury in environmental samples.

2. Experimental

Unless otherwise noted, solvents and chemicals were used as received from Sigma–Aldrich (St. Louis, MO). The HN-Eu₂O₃ was



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synthesized by a simple one-step template-free procedure in aqueous solution. In a typical experiment, 0.10g of europium acetate $(Eu(Ac)_3)$ was dissolved into 25 ml of water, The solution was stirred with a magnetic stir bar and heated to 90 °C. A 1.0 ml aliquot of 20 mg/ml hexamethylenetetramine (HMT) was added dropwise. The clear mixture turned cloudy in 30 min, became opaque again after another 30 min, and some white precipitate was found at the bottom of the flask. Ninety minutes after the addition of HMT, the precipitate was collected and purified by multiple centrifugation–dispersion cycles with alcohol. This precursor was then calcinated at 500 °C for 3 h to obtain highly crystalline HN-Eu₂O₃. TTA capping was achieved by treating the HN-Eu₂O₃ with a saturated TTA solution in ethanol.

Transmission electron microscopic (TEM) experiments were performed on a FEI Tecnai G2 F20 electron microscope operated at 200 kV. Field-emission scanning electron microscopic (FESEM) experiments were conducted on a JEOL JSM-7400F electron microscope. UV-vis and fluorescence spectra were recorded on a V-570 UV/VIS/NIR spectrophotometer (JASCO Corp., Japan) and a Fluorolog[®]-3 spectrofluorometer (Jobin Yvon Inc, Edison, NJ), respectively. The quantum yield of the cHN-Eu₂O₃ was estimated by measuring integrated fluorescence intensities of the cHN-Eu₂O₃ sample and the reference (Eu(TTA)₃(phen) in DMF, QY = 36.5%) under 380 nm excitation [18]. Fluorescence images of the cHN-Eu₂O₃ were acquired with an Olympus Inverted Microscope. Brunauer–Emmett–Teller (BET) surface area was obtained using a Micromeritics ASAP 2020 M system.

Mercury analysis: for each sample three replicates of approximately 0.50 g were transferred to 25 ml polypropylene test tubes. To each replicate 5.0 ml of 3:1 (w/w) H₂SO₄:HNO₃ was added and the mixture was heated to 80 °C for 120 min. Then 10.0 ml of 6 M HCl and 3.0 ml of 0.1 M KBrO₃/KBr solution were added. After mixing, the sample was heated to 60 °C in a water bath overnight. The sample solution was yellow in color and clear without any precipitate. Next, 1.0 M hydroxylamine hydrochloride was slowly added until the solution becomes colorless. For spiked samples, Hg²⁺ calibration standards were added to the polypropylene test tubes before digestion. Water samples were filtered through a P10 glass filter prior to digestion. To a suitable aliquot of sample solution added $1.0 \,\mu$ g/ml cHN-Eu₂O₃. After equilibrating at room temperature for 30 min under stirring, the fluorescence of each mixture was measured. For control experiments, the same procedure was applied, except that blank 0.10 M NaNO3 solution used instead of the mercury sample solution.

3. Results and discussion

Fig. 1a is a FESEM image of the HN-Eu₂O₃, showing uniform spherical particles with a diameter of ~1 μ m. Energy dispersive X-ray analysis indicated that the particles are essentially pure Eu₂O₃. A high-magnification FESEM image (Fig. 1, insert) revealed that the spherical particles are actually built from several dozen nanosheets with smooth surfaces. These nanosheets, with a thickness of 2–5 nm, are interconnected to each other to form hierarchical nanostructures. Crystalline grains in TEM micrographs (Fig. 1b) and a series of diffraction rings in a selected-area electron diffraction experiment indicated that the HN-Eu₂O₃ is essentially polycrystalline. Depending on the particle size, BET surface area was found to be from 180 to 230 m²/g, which is comparable to those of other metal oxide-based HN-particles [8]. Smaller HN-Eu₂O₃ spheres down to 200 nm can be prepared with higher Eu(Ac)₃ concentrations while keeping the Eu(Ac)₃/HMT ratio unchanged.

Fig. 2a shows UV–vis absorption spectra of the HN-Eu₂O₃ before and after the TTA capping. For comparison, the UV–vis spectrum of TTA in water is also given in Fig. 2a (trace 3). The spectrum of

Fig. 1. (a) Representative FESEM images of the $HN-Eu_2O_3$ at low magnification (scale bar 20 μ m) and at high magnification (insert, scale bar 0.5 μ m). (b) A high resolution TEM image of $HN-Eu_2O_3$ and the corresponding SAED pattern (insert).

the HN-Eu₂O₃ before capping was more or less characteristic of the spectrum of nanoparticulate Eu₂O₃: a rather broad absorption band stretches over 350 nm (Fig. 2a, trace 2). The principal features were two absorption peaks at 230 and 270 nm, which correspond to transitions from the ${}^{7}F_{0}$ ground state to charge transfer states due to europium-oxygen interaction. [19]. The long absorption tail is due to scattering by the HN-Eu₂O₃ spheres [20]. The spectrum of the cHN-Eu₂O₃ appeared not as a simple superposition of TTA and the HN-Eu₂O₃, but as a superposition of the HN-Eu₂O₃ and an europium-TTA complex (Fig. 2a, trace 1) with a main absorption peak in the 300-400 nm region [21], indicating that TTA is co-ordinatively grafted onto the HN-Eu₂O₃. Due to the hierarchical nature of the nanostructure and the ultrathin nanosheet building blocks, a very large portion of Eu₂O₃ is located on the surface, capping with TTA inevitably alerts the chemical environment for most of Eu_2O_3 .

The excited states of Eu^{3+} are known to be very sensitive to water. Water molecules in the first co-ordination sphere of Eu^{3+} provide an efficient route for radiationless deactivation, which is particularly efficient in co-ordinatively unsaturated systems [22]. As expected, no Eu^{3+} emission was visible in the as-obtained HN- Eu_2O_3 in water under UV-irradiation (Fig. 2c, insert). On the other hand, upon capping, the water molecules on the HN- Eu_2O_3 surface were displaced by TTA, leading to the formation of a TTA outlayer. This TTA outlayer acts not only as a water repelling layer, but also as an efficient light-harvesting layer by its well-known antenna effect [23]. The substitution of H₂O by TTA on the HN- Eu_2O_3 surface effectively removes the nonradiative decays caused by weak couplings of the Eu^{3+} excited states with the OH oscilla-





Fig. 2. (a) UV absorption spectra of (1) 10 µg/ml the cHN-Eu₂O₃, (2) 10 µg/ml the HN-Eu₂O₃ and (3) 10 µg/ml TTA in water. (b) (1) Emission and (1') excitation spectra of the cHN-Eu₂O₃; (2) emission and (2') excitation spectra of Eu₂O₃ nanoparticles, and (c) a typical fluorescence image of the cHN-Eu₂O₃ taken under UV illumination. Insert: photographs, taken under a UV lamp (left) before and (right) after the TTA capping.

tors, resulting in a dramatic increase in the fluorescence intensities. As illustrated in Fig. 2c, the red fluorescence of cHN-Eu₂O₃ was so intensive that individual HN-Eu₂O₃ spheres were clearly visible in the solid state under a fluorescence microscope. The red emission from cHN-Eu₂O₃ was also readily visible to the naked eye upon excitation with a UV lamp (Fig. 2c, insert). Up illumination, the capping layer is excited and then decays rapidly to its lowest triplet state. And more importantly, the majority of Eu₂O₃ is able to receive the energy transferred from the excited TTA due again to the unique hierarchical structure and ultrathin nanosheet building blocks. In addition, the typical red emission of the cHN-Eu₂O₃ indicates that the Eu³⁺ emitting center does not occupy a center of symmetry. If the Eu³⁺ center were at the center of symmetry, then the emission should appear orange [24]. This intensive Eu³⁺ emission was further investigated by fluorescence spectrophotometry. The emission spectrum of the cHN-Eu₂O₃ (Fig. 2b, trace 1) displayed the characteristic ${}^{5}D_{0}-{}^{7}F_{I}$ (J ${}^{1}\!\!\!/_{4}$ 0–4) transitions of Eu $^{3+}$ under excitation from ${}^{7}F_{0}$ to ${}^{5}L_{6}$ level [16], or under ligand excitation. The hypersensitive ${}^{5}D_{0}-{}^{7}F_{2}$ transition is predominant in all cases, suggesting that Eu³⁺ ions lie in a non-centrosymmetrical environment [16]. In the spectrum, there was no evidence of emissions from the triplet state of TTA, implying that there is an efficient energy transfer from the triplet state of TTA to the HN-Eu₂O₃. It is well-known that the magnetic-dipole transitions ${}^{5}D_{0}-{}^{7}F_{1}$ are nearly independent of the ligand field, and therefore can be used as an internal standard to account for ligand difference [25]. The electric-dipole transitions ${}^{5}D_{0}-{}^{7}F_{J=2,4,6}$, the so-called hypersensitive transitions, are sensitive to the symmetry of the co-ordination sphere. The intensity ratio of the magnetic-dipole transition to the electric-dipole transition in the europium complex measures the symmetry of the co-ordination sphere [23,26]. The intensity ratios of the ${}^{5}D_{0}-{}^{7}F_{2}$ transition to the ${}^{5}D_{0}-{}^{7}F_{1}$ transition in pure Eu₂O₃ nanoparticles (30-50 nm, from Nanostructured & Amorphous Materials, Inc. Houston, TX 77084) and the cHN-Eu₂O₃ were determined to be 11.5 and 13.5, respectively. These results show that Eu³⁺ ions in the cHN-Eu₂O₃ exhibit different local environments because of the influence of the surrounding TTA molecules. As seen in Fig. 2b traces 1 and 2, the fluorescence intensity the cHN-Eu₂O₃ was more than 1000 times higher than that of the commercial Eu₂O₃ nanoparticles (equivalent weight) and ~28 higher than that of the commercial Eu_2O_3 nanoparticles coated with TTA. The emission life time and quantum yield of the cHN- Eu_2O_3 were found to be ~ 1.0 ms and $\sim 68\%$, respectively.

The excitation spectrum (Fig. 2b, trace 1') of the cHN-Eu₂O₃ displays a predominant TTA ligand excitation band at 390 nm, as a result of the antenna effect [24], while the excitation spectrum of the Eu₂O₃ nanoparticles was given as a reference (Fig. 2b, trace 2'). Moreover, the absorption spectrum revealed that the excitation bands are displaced to lower energies when compared to the absorption bands of the free ligand (Fig. 2a, trace 3), indicating the co-ordinative nature of TTA with Eu³⁺ ion. It was clearly observable that the cHN-Eu₂O₃ is strongly excitable over all the low-energy ultraviolet region of the spectrum. In addition, in the excitation spectrum of the HN-Eu₂O₃, the ⁷F–⁵D inner-shell excitation lines at 465 nm persists, suggesting that the f–f inner-shell transitions are largely unaffected and there is very little of quenching through nonradiative energy transfer from the excited state to the TTA capping layer.

Potential applications of hierarchically nanostructured materials include catalysis, energy conversion and storage, environmental abatement, and sensors. The highly intensive fluorescence, in combination with the networks formed by the interconnected nanosheets, producing multiscale porosity, high surface-to-bulk ratio, and ultralarge surface area, makes the cHN-Eu₂O₃ a good candidate for heavy metals removal and possibly an ideal sensing material for probing the heavy metals simultaneously. Indeed, it was observed that traces of mercury effectively quench the fluorescence of the cHN-Eu₂O_{3.} leading to potential applications in the determination of mercury. The ability of the cHN-Eu₂O₃ to detect mercury was then investigated by adding successive aliquots of aqueous solutions of mercury to aqueous suspensions of the cHN-Eu₂O₃ and monitoring the changes in fluorescence intensity (Fig. 3a). The fluorescence of the cHN-Eu₂O₃ was completely quenched by mercury at concentrations higher than 20 ppm. Quenching efficiencies were fit to the Stern-Volmer equation, $I_0/I = K_{sv}[A] + 1$ [27], where *I* is the fluorescence intensity, I_0 is the intensity at [A] = 0, and K_{SV} is the Stern–Volmer constant. A plot of mercury quenching efficiency was linear up to 10 ppm, at which point there was a gradual decrease in quenching efficiency (Fig. 3b). This nonlinear relationship observed may indicate saturation of the surface binding sites. The selectivity of the cHN-Eu₂O₃ was tested by performing quenching experiments at 1.0 ppm Hg²⁺ in the presence of common metal ions. As shown in Fig. 4, the fluorescence intensity of the cHN-Eu₂O₃ was little affected by the presence of 10 ppm of Ca²⁺, Fe²⁺, Fe³⁺, Cu²⁺, Mg²⁺, Ni²⁺, Co²⁺, Al³⁺, Zn²⁺, and Mn²⁺. However, it was observed that the fluorescence intensity was reduced by 15% in the presence of 10 ppm Pb²⁺ and it was almost completely quenched when Pb²⁺ concentration was increased to 50 ppm, implying some interference from Pb²⁺. Better selectivity was observed when the determination of mercury was carried out in the presence of 50 µM phytic acid (masking agent), due to the fact that phytic acid forms stable complexes with many metal ions, including Fe²⁺, Cu²⁺, Ni²⁺, Co²⁺, Zn²⁺, and Pb²⁺ [28]. For example, the cHN-Eu₂O₃ produced ~8-fold higher selectivity toward Hg²⁺ ions over the other metal ions when mercury was analyzed in solutions containing 50 µM phytic acid.

The proposed method was applied to the determination of mercury by the single standard addition method. The samples were digested and separated using the procedure recommended by Marczenko [29]. A 50 mM NaNO₃ solution was used as the medium for mercury sample preparation. As seen in Table 1, good agreement between the proposed method and atomic adsorption spectrometry (AAS) was obtained. A relative standard deviation of 10% was obtained for six determinations of 100 ppb mercury. The recoveries obtained (94–103%) were also good enough for practical use.



Fig. 3. (a) Changes in the emission of the cHN-Eu₂O₃ (excited at 380 nm) in the presence of increasing amounts of mercury in 2.5 ppm increments and (b) a plot of the relative intensity vs. mercury added.



Fig. 4. Effect of common metal ions on the determination of 1.0 ppm mercury. Experimental conditions are as for Fig. 3.

Table 1Results of mercury analysis (n = 6).

Sample	Found (this method) $(\mu g/g)$	Found (AAS) ($\mu g/g$)
Hair	0.33	0.29
Shell fish	0.130	0.120
Fish	0.232	0.235
Waste water	0.093	0.090

4. Conclusion

In summary, we have developed a fluorescence quenching procedure for sensitive determination of mercury using the cHN-Eu₂O₃. The HN-Eu₂O₃ was synthesized by a simple one-step template-free process. Capping the HN-Eu₂O₃ with TTA resulted in an enormous enhancement in fluorescence, which is much more intensive than that of commercial Eu₂O₃ nanoparticles under the same conditions. The excellent fluorescence properties of the cHN-Eu₂O₃ are related to their special structural uniqueness and imply potential applications in catalysis, energy conversion, and sensing. The cHN-Eu₂O₃ was then utilized to determine trace amounts of mercury in various samples. Further studies will focus on extending to the synthesis of hierarchical nanostructures and their applications.

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